

Enhanced Photocatalytic Performance of Nanocrystalline TiO₂ Membrane by Both Slow Photons and Stop-band Reflection of Photonic Crystals

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An nc-TiO₂/SnO₂ inverse opal composite membrane was fabricated, the photo-activity of which was significantly enhanced by utilizing both slow photons and stop-band reflection of the photonic crystal layer. The materials of the photonic crystal layer must be transparent in the area of adsorption edge of the nc-TiO₂, so that SnO₂, having much greater electronic band gap than TiO₂, was used for the materials of the photonic crystal layer. The photonic band-gap of the SnO₂ photonic crystal was designed at the semiconductor band gap of TiO₂ to harvest slow photons in the interface between the SnO₂ layer and the TiO₂ layer. The two layer structure makes it possible to couple the stop-band reflectivity of the photonic layer to the photocatalyst. Composite membranes can improve solar energy harvesting and substantially improve photocatalysts for photolysis and photochemical degradation of environmental pollutants. © 2011 American Institute of Chemical Engineers AICHE J, 58: 568–572, 2012

Keywords: nanocrystalline TiO₂, slow photons, photonic crystals, SnO₂, inverse opal, stop-band reflection, photo-catalyst

Introduction

Titanium oxide (TiO₂) is widely used as a photocatalyst for environmental cleanup and photolysis of water. As a photocatalysts, it is chemically stable in reactive environments, is nontoxic and is available at low cost.^{1,2} However, the solar energy harvesting efficiency of TiO₂ is low. Its moderate electronic band-gap ($E_g = 3.2\text{eV}$) limits the useful range of the solar spectrum that can excite TiO₂ to a few percent. The lifetimes of the photo-excited states are short, and are frequently quenched before reaching the surface where reactants are adsorbed. Despite its shortcomings TiO₂ has been the photocatalyst of choice for scientific studies since the discovery of photocatalytic splitting of water on TiO₂ electrodes in 1972.³ Extensive efforts have attempted to enhance the photocatalytic efficiency of TiO₂ by chemists, physicists, and chemical engineers; success has been limited, and there are few commercial applications of photocatalysis. We seek to improve the efficiency of TiO₂ photocatalysts by increasing light absorption in thin films of nanocrystalline thin films.

Energy harvesting efficiency by TiO₂ may be enhanced by either reducing the band-gap to more favorably match the solar spectrum, or by increasing the transport of excitons to the surface for chemical reaction.^{4,5} Solar absorption may be

enhanced by decreasing the band-gap to access a larger portion of the solar spectrum. TiO₂ can be doped with impurities to reduce the band gap energy,^{6,7} or dyes may be added to absorb photons at lower energies.^{1,8} However, increasing the wavelength response range sometimes decreases the photocatalytic efficiency of TiO₂ because of the lower redox potential of electrons and holes created by photoexcitation. The most effective approach to improve photocatalytic activity of TiO₂ is to enhance absorption of thin films without chemical compositional changes.

Thin films are necessary to permit electrons and holes to migrate to the interface between the TiO₂ and reactant before quenching. Unfortunately, the absorptivity of TiO₂ is small and light passes through thin films of TiO₂ with only a small fraction of the sub-band-gap light absorbed. If the light could be trapped in thin TiO₂ films until it is absorbed, energy harvesting would be enhanced.

Recently, Mallouk and coworkers^{9,10} reported a 26% increase of the short current photocurrent of a dye-sensitized solar cell (DSSC) constructed of a bilayer structure of a thick nanocrystalline TiO₂ (nc-TiO₂) layer on top of a TiO₂ inverse opal. The photonic stop band of the inverse opal was 610 nm, close to the absorbance edge of the dye used in the DSSC. They attributed the increased current to light absorption by the dye. Subsequent theoretical calculations by Mihi and Míguez¹¹ indicated that the increased efficiency could be explained by the inverse opal acting as a dielectric mirror, which increased the effective pathlength of light through nc-TiO₂ layer, permitting greater absorption.

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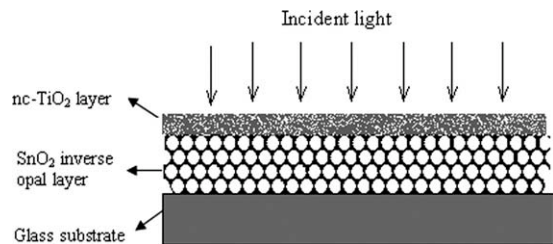


Figure 1. The structure of the composite photocatalytic membrane.

Ozin and coworkers demonstrated the use of TiO_2 photonic crystals to improve photocatalytic activity. They fashioned the photonic crystal to have a photonic band-gap close to the electronic band-gap of TiO_2 . The photonic crystal slowed down the phase velocity of light, creating “slow-photons” that are more effectively absorbed by the TiO_2 nanocrystals.^{12,13} They absorbed methylene blue into their photonic crystals and showed that photodegradation was enhanced by matching the photonic crystal band-gap to the electronic band-gap of TiO_2 . This report by Ozin and coworkers appears to be the first article specifically addressing photocatalytic activity enhancement by photonic crystals.

Photonic crystals have a periodic dielectric contrast at the length scale of the wavelength of light. Bragg diffraction forbids light with certain wavelength to propagate through the material. This gives rise to stop-band reflection, and the range of energies that is reflected back depends on the periodicity and dielectric contrast of the photonic crystal. At the frequency edges of these photonic stop bands, photons propagate with strongly reduced group velocity in the periodic photonic structures, giving rise to “slow photons” that have a long-effective path length through the photonic crystal. These slow photons can be observed in disfigurement of the periodic photonic structures at energies just above and below the photonic stop band¹⁴

Ozin and coworkers used the slow photons of the photonic band gap to amplify the photoactivity of TiO_2 within the photonic crystal, but the reflected light of stop-band reflection was not used. To use both the slow photons of the photonic band gap and the band gap reflection, we constructed a composite photocatalytic membrane by coupling a SnO_2 photonic crystal layer to a nc- TiO_2 catalyst membrane. The photonic band gap of the SnO_2 layer was designed to match the electronic stop band of TiO_2 semiconductor. The structure of the composite membrane is shown in Figure 1. By separating the functionality of the photonic crystal and the photocatalyst, it also permits greater flexibility in choosing materials for the two.

When sunlight hits on the nc- TiO_2 layer, a fraction of the light with shorter wavelengths than the electronic stop band of TiO_2 is absorbed creating electron hole pairs that migrate to the surface of the nc- TiO_2 catalyst membrane. Light with wavelengths longer than the electronic stop band of TiO_2 passes through the nc- TiO_2 layer. Normally, only a small fraction of the short wavelength light is absorbed in the thin TiO_2 film. By using a photonic crystal as a dielectric mirror, it is possible to increase the light path through the thin TiO_2 layer and enhance the absorption of light.

The material in the photonic crystal should be transparent to light at energies above the electronic stop band of the TiO_2 to minimize light attenuation within the photonic crystal. It should also have a high dielectric contrast at

higher energies to reflect the light at energies above TiO_2 's band gap. The photonic crystal layer acts as a dielectric mirror for the slow photons. Light is absorbed by evanescent waves upon reflection at the interface between the photonic crystal and TiO_2 . The slow photons in the photonic crystal create a long effective path length for the evanescent waves so most of the light capable of exciting TiO_2 does so.

The photonic crystal is a multiple internal reflection crystal with an absorbing film coating one surface of the crystal. Light is absorbed by the nc- TiO_2 layer. There is no alteration of the band gap of the TiO_2 so the redox potential of electrons and holes is retained. The photocatalytic activity of the composite photocatalyst-photonic crystal membrane should be not only higher than that of a plain nc- TiO_2 membrane, but also higher than that of TiO_2 fashioned as inverse opals as reported by Ozin and coworkers,^{12,13} because the composite membrane proposed in this article can use both slow photons and stop-band reflection produced by photonic crystals.

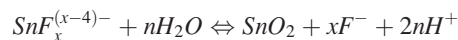
We chose tin oxide (SnO_2) for the photonic crystal, because SnO_2 has a larger electronic band gap ($E_g = 3.8$ eV) than TiO_2 and is almost transparent in the electronic stop band of TiO_2 . SnO_2 is also chemically stable and inexpensive making it a near ideal material of choice for the photonic layer. As it is difficult to prepare monodispersed SnO_2 spheres, the photon crystals were constructed as SnO_2 inverse opal structure.

Experimental

Preparation of SnO_2 liquid-phase deposition (LPD) solution

SnF_2 was dissolved into DI water. Drop-wise addition of H_2O_2 resulted in precipitation of $\text{SnO}_2 \cdot n\text{H}_2\text{O}$. The precipitate was washed with DI water repeatedly, dried at room temperature, and then dissolved into 40% hydrofluoric acid. The SnO_2 LPD solution was prepared by pouring 1.5 mol/L H_3BO_3 solution into the SnF_4 -HF solution at the final concentrations of the 0.025 mol/L SnF_4 -HF and 0.2 mol/L H_3BO_3 . The H_3BO_3 was used as an F^- scavenger.

SnO_2 was formed through the ligand-exchange hydrolysis of metalfluoro complexes. Boric acid is added to the stannum-fluoro complex solution to react with the F^- ions forming BF_4^- , shifting the hydrolysis equilibrium to form SnO_2 , as follows.



Preparation of SnO_2 inverse opal layer

Quartz substrates were immersed vertically into a dispersion solution of monodispersed polystyrene (PS) particles, which were prepared through emulsion polymerization method.¹⁵ The dispersion solution of PS with the quartz substrates was maintained at 50°C to slowly evaporate water and a PS opal layers were deposited on the quartz substrates. The quartz substrates with PS opal layers were immersed vertically into the SnO_2 LPD solution at 30°C and the SnO_2 precipitated and filled into voids of the PS opal layer. After maintaining in the solution for about 10 h, the quartz substrates with PS opal/ SnO_2 were removed from the SnO_2 LPD solution, washed with DI water, and dried at room temperature. The SnO_2 inverse opal layer was obtained by calcination of the quartz substrates in air at 450°C for 2 h.

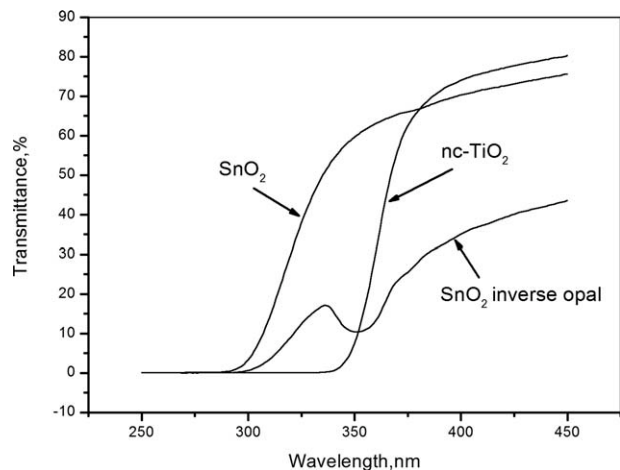


Figure 2. The UV-vis transmittance spectra of nc-TiO₂ and SnO₂ and SnO₂ inverse opal membranes.

Preparation of nc-TiO₂/SnO₂ inverse opal composite membranes

The nc-TiO₂ was prepared through sol-gel method. A 40-mL of mixture of titanium isopropoxide and iso-propylenol (25:15 volume) was added into 150-mL dilute aqueous solution of HNO₃, and TiO₂ gel was precipitated from the reaction mixture by the hydrolysis of the titanium isopropoxide. The solution containing the TiO₂ precipitate was heated to 80°C and kept at 80°C for 8 h, then transferred into an autoclave, heated to 220°C and hydro-treated for 12 h. An nc-TiO₂ dispersion was obtained by cooling down the TiO₂ dispersion. Before spin coating the dispersions, they were sonicated for 10 min. The nc-TiO₂ was coated onto the SnO₂ inverse opal membrane through spin-coating, and the thickness of the nc-TiO₂ layer was controlled by spin speed rate and the solid content of the nc-TiO₂ sol solution. The nc-TiO₂/SnO₂ inverse opal composite membranes were obtained through calcining the membranes at 450°C for 2 h.

Photodegradation performance test

The photodegradation of methyl orange (MO) was used as the probe reaction to test the photocatalytic activity of the

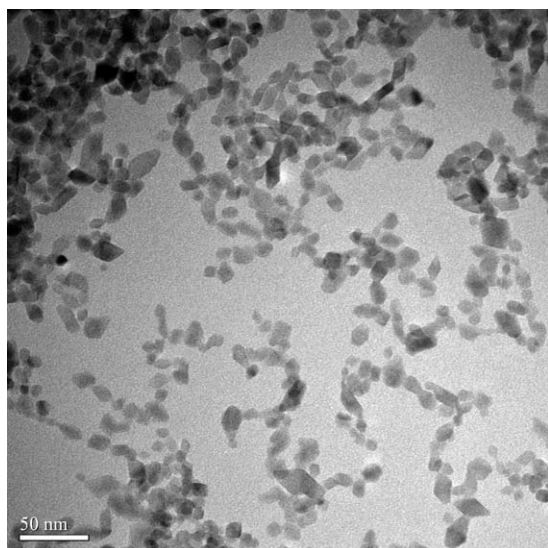


Figure 3. The TEM of the prepared nc-TiO₂ particles.

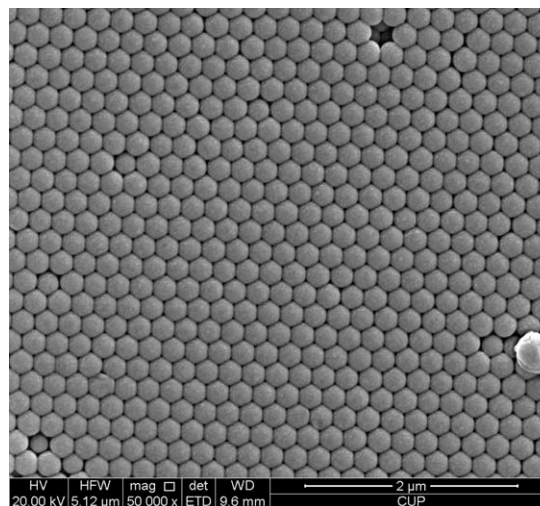


Figure 4. The SEM of the top view of the PS opal layer deposited on a quartz substrate.

membranes. The concentration of the MO was 10 mg/L. A photocatalyst membrane of 25 × 10 mm was fixed in the center of a 1 mm × 1 mm × 4.5 cm quartz ultraviolet (UV) cell and 3-mL MO solution was put into the UV cell. The UV cell was then placed into a light tight box and irradiated by a 150W Xe lamp; the intensity at the UV cell was 3.5 mW/cm². The concentration change of the MO was followed through recording the UV-Vis spectra of the MO solution at different time down the reaction using a UV-vis spectrometer. The reaction conditions were the same for all experiments.

Results and Discussions

The UV-vis absorption spectrum SnO₂ and TiO₂ layers are shown in Figure 2; light between 300 and 350 nm is able to propagate through the SnO₂ photonic crystal creating slow photons that can be absorbed by TiO₂. The thin film of TiO₂ was from nanocrystalline anatase particles. X-ray diffraction (XRD) of the nc-TiO₂ particles indicates that the prepared TiO₂ particles were anatase. The transmission electron microscopy (TEM) of the nc-TiO₂ particles as shown in Figure 3 shows that these particles were of 5~20 nm.

SEM images of the prepared PS opal layer and the prepared SnO₂ inverse opal membrane, as shown in Figures 4 and 5 respectively, reveal that the PS spheres were highly ordered packed as the opal layer and the its replica-- SnO₂ inverse opal membrane, prepared by filling SnO₂ into the PS opal and then removing PS spheres, inherited the ordered structure of the PS opal.

An SEM image of the cross-section of the prepared nc-TiO₂/SnO₂ inverse opal composite membrane is shown in Figure 6, indicating that the thicknesses of the SnO₂ photonic layer and the nc-TiO₂ layer are ~1.2 and ~0.55 μm, respectively.

The photonic band gap of the SnO₂ inverse opal was obtained at normal incidence to compare to the electronic band gap of TiO₂. A sample of the SnO₂/opal PS membrane was calcined at 450°C before coating the nc-TiO₂ to obtain a SnO₂ inverse opal. The UV-Vis transmittance spectrum of the SnO₂ photonic crystal is compared with the UV-vis absorption spectra of TiO₂ and SnO₂ in Figure 2. The

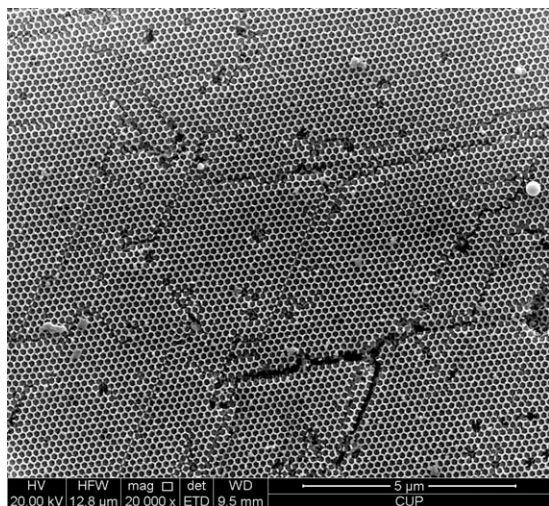


Figure 5. The SEM of the top view of the SnO₂ inverse opal composite membrane.

photonic band gap is evident at 335~370 nm, overlapping with the electronic band gap of the anatase TiO₂.

Photocatalytic activity of the membranes was tested using decomposition of methyl orange (MO) in water solution as the probe reaction. The MO-containing solution was contained in a UV-cell, in which a TiO₂ photocatalyst substrate was suspended. The UV cell was irradiated by a Xe lamp and in a light tight box. The UV cell was removed periodically and the UV-vis spectrum recorded to determine the MO concentration. The photocatalytic decomposition of the MO follows first-order decay rate law. The decay rate constant was obtained from the logarithmic plot of the relative concentration of the MO as a function of time. It was previously reported that the SnO₂ added into the TiO₂ as a doping agent can increase the photoactivity of the TiO₂,^{16,17} because the conduction band of SnO₂ is lower than that of the TiO₂, the former acts as a sink for the photo-generated electrons thereby making charge separation more efficient and improving the photocatalytic activity. To elucidate the photoactivity produced by the slow photons, the photocatalytic activity of an nc-TiO₂ membrane supported on quartz and an nc-TiO₂

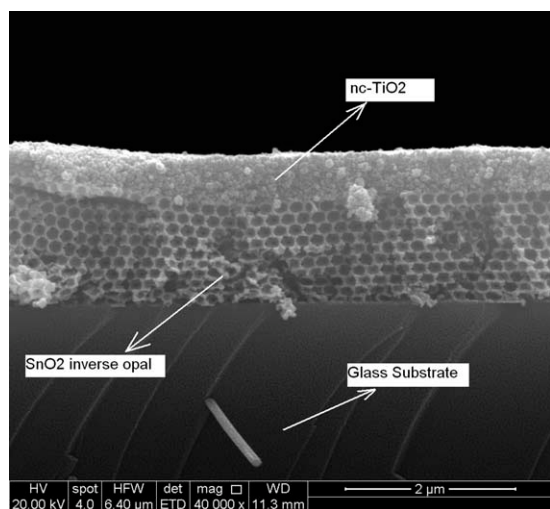


Figure 6. The SEM of the cross-section of the nc-TiO₂/SnO₂ inverse opal composite membrane.

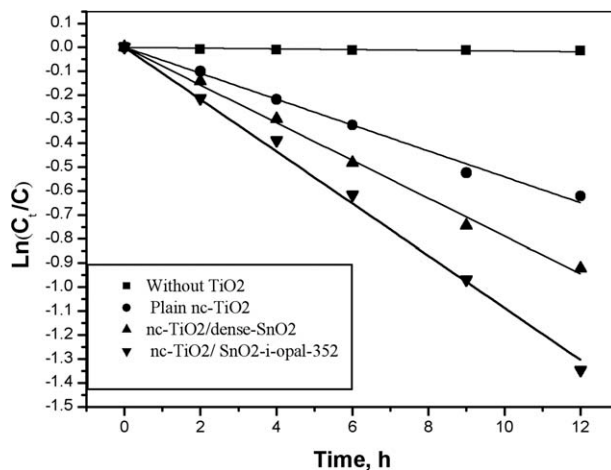


Figure 7. Photocatalytic degradation kinetic curves of MO.

(Reaction condition: 20°C, 3 mL 10 ppm MO in 1 cm × 1 cm × 4.5 cm UV cell, 3.5 mW/cm² radiation of Xe lamp, and 1 cm × 2.5 cm area of the photocatalyst).

membrane supported on a dense SnO₂ layer were compared with the nc-TiO₂ layer supported on the SnO₂ photonic crystal. For convenience, the composite membrane of nc-TiO₂/SnO₂ inverse opal, the membrane of nc-TiO₂ layer supported on a dense SnO₂ layer and the membrane of nc-TiO₂ layer supported on quartz are designated as nc-TiO₂/SnO₂-i-opal-352, nc-TiO₂/dense-SnO₂ and plain nc-TiO₂, respectively. Figure 7 shows the logarithmic plot of the concentration of the MO as a function of time for variety of membranes. The photocatalytic activity of these membranes is in the order of nc-TiO₂/SnO₂-i-opal-352 > nc-TiO₂/dense-SnO₂ > plain nc-TiO₂; the decay rate constants of MO on these membranes are 0.109 h⁻¹, 0.0787 h⁻¹, and 0.0557 h⁻¹, respectively. The increased decomposition rate indicates greater absorption of the incident radiation. The greater photocatalytic activity of the nc-TiO₂/dense-SnO₂ membrane compared to a plain nc-TiO₂ membrane is the result of the SnO₂ acting as a sink for the photo-generated electrons thereby making charge separation more efficient. The light absorption is further enhanced by employing a SnO₂ photonic crystal with “slow photons.”

The reason why the photoactivity of the nc-TiO₂/SnO₂-i-opal-352 composite membrane is higher than nc-TiO₂/dense-SnO₂ membrane is that the former makes use of the slow photons at the stop band reflection produced by the SnO₂ photonic crystal layer. The slow photons are produced at the interface between the nc-TiO₂ layer and the SnO₂ photonic crystal layer enhancing absorption and the photoactivity of the nc-TiO₂ layer.

The key difference in light energy utilization between Ozin's membranes and our composite membranes is that more photons near the external surface of the composite membrane are utilized. The composite structure we introduced also exploits both slow photons and stop band reflection, while the TiO₂ inverse opal membranes reported by Ozin and coworkers used only slow photons and the stop band reflection energy was wasted. Furthermore, Ozin used the internal surface area of the TiO₂ inverse opal and all their reactant was adsorbed on the TiO₂ surface before irradiation. They had a stoichiometric reaction. In contrast, our membranes had sustained catalytic cycles, the MO adsorbed from solution, reacted, products desorbed and the catalytic site was restored.

The coupling of photonic crystals with photocatalysts can offer a variety of novel fabrication possibilities.

Conclusions

We fabricated an nc-TiO₂/SnO₂ inverse opal composite membrane, the photoactivity of which was significantly enhanced by utilizing both slow photons and stop-band reflection of the photonic crystal layer. The materials of the photonic crystal layer must be transparent in the area of adsorption edge of the nc-TiO₂, so that SnO₂, having much greater electronic band gap than TiO₂, was used for the materials of the photonic crystal layer. The photonic band-gap of the SnO₂ photonic crystal was designed at the semiconductor band gap of TiO₂ to harvest slow photons in the interface between the SnO₂ layer and the TiO₂ layer. The two-layer structure makes it possible to couple the stop-band reflectivity of the photonic layer to the photocatalyst. Composite membranes can improve solar energy harvesting and substantially improve photocatalysts for photolysis and photochemical degradation of environmental pollutants.

Acknowledgments

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Literature Cited

- Hoffmann MR, Martin ST, Choi W, Bahnemann DW. Environmental applications of semiconductor photocatalysis. *Chem Rev.* 1995;95:69–96.
- Liu S, Liu H. *Applications and Fundamentals of Photocatalysis and Electrocatalysis*, 1st ed. Beijing: Chemical Industry Press, 2006:74–75.
- Fujishima A, Honda K. Electrochemical photolysis of water at a semiconductor electrode. *Nature.* 1972;238:37–38.
- Choi W, Termin A, Hoffmann MR. The role of metal ion dopants in quantum-sized TiO₂: correlation between photo-reactivity and charge carrier recombination dynamics. *J Phys Chem.* 1994;98:13669–13679.
- Bamwenda GR, Tsubota S, Nakamura T, Haruta M. Photoassisted hydrogen production from a water-ethanol solution: a comparison of activities of Au-TiO₂ and Pt-TiO₂. *J Photochem Photobiol A.* 1995;89:177–189.
- Asahi R, Morikawa T, Ohwaki T, Aoki K, Taga Y. Visible-light photocatalysis in nitrogen-doped titanium dioxide. *Science.* 2001;293:269–271.
- Sathish M, Viswanathan B, Viswanath RP, and Gopinath CS. Synthesis, characterization, electronic structure and photocatalytic activity of nitrogen-doped TiO₂ nanocatalyst. *Chem Mater.* 2005;17:6349–6353.
- Zhang W, He M, Li X. Progress of Nano- TiO₂ modification in visible light. *Sci Technol Rev.* 2007;25:72–78.
- Nishimura S, Abrams N, Lewis BA, Halaoui LI, Mallouk TE, Benkstein KD, van de Lagemaat J, Frank AJ. Standing wave enhancement of red absorbance and photocurrent in dye-sensitized titanium dioxide photoelectrodes coupled to photonic crystals. *J Am Chem Soc.* 2003;125:6306–6310.
- Halaoui LI, Abrams NM, and Mallouk TE. Increasing the conversion efficiency of dye sensitized TiO₂ photoelectrochemical cells by coupling to photonic crystals. *J Phys Chem B.* 2005;109:6334–6342.
- Miñi A, and Míguez H. Origin of light-harvesting enhancement in colloidal-photonic -crystal-based dye-sensitized solar cells. *J Phys Chem B.* 2005;109:15968–15976.
- Chen JIL, von Freymann G, Choi SY, Kitaev V, Ozin GA. Amplified photochemistry with slow photons. *Adv Mater.* 2006;18:1915–1919.
- Chen JIL, von Freymann G, Choi SY, Kitaev V, Ozin GA. Slow photons in TiO₂ inverse opals: optical amplification and effect of disorder on the photo-catalytic efficiency. *Proc SPIE.* 2007;6650:6650W-1–6650W-7.
- Hore S, Nitz P, Vetter C, Prahl C, Niggemann M, and Kern R. Scattering spherical voids in nanocrystalline TiO₂ – enhancement of efficiency in dye-sensitized solar cells. *Chem Commun.* 2005; 2011–2013.
- Chen SL, Xu KQ, Dong P. Preparation of three-dimensionally ordered inorganic/organic bi-continuous composite proton conducting membranes. *Chem Mater.* 2005;17:5880–5883.
- Hattori A, Tokihisa Y, Tada H, Ito S. Acceleration of oxidations and retardation of reductions in photocatalysis of a TiO₂/SnO₂ bilayer-type. *J Electrochem Soc.* 2000;147:2279–2283.
- Vinodgopal K, Bedja I, Kamat PV. Nanostructured semiconductor films for photocatalysis photoelectrochemical behavior of SnO₂/TiO₂ composite systems and its role in photocatalytic degradation of a textile azo dye. *Chem Mater.* 1996;8:2180–2187.

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